

Does High-Entropy Oxide (CrMnFeCoNi)₃O₄ Catalyst Require Support to Improve Performance in CO₂ Hydrogenation?

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STEM images of the HEO/TiO₂ and HEO/BN samples are collected in Figure S1.

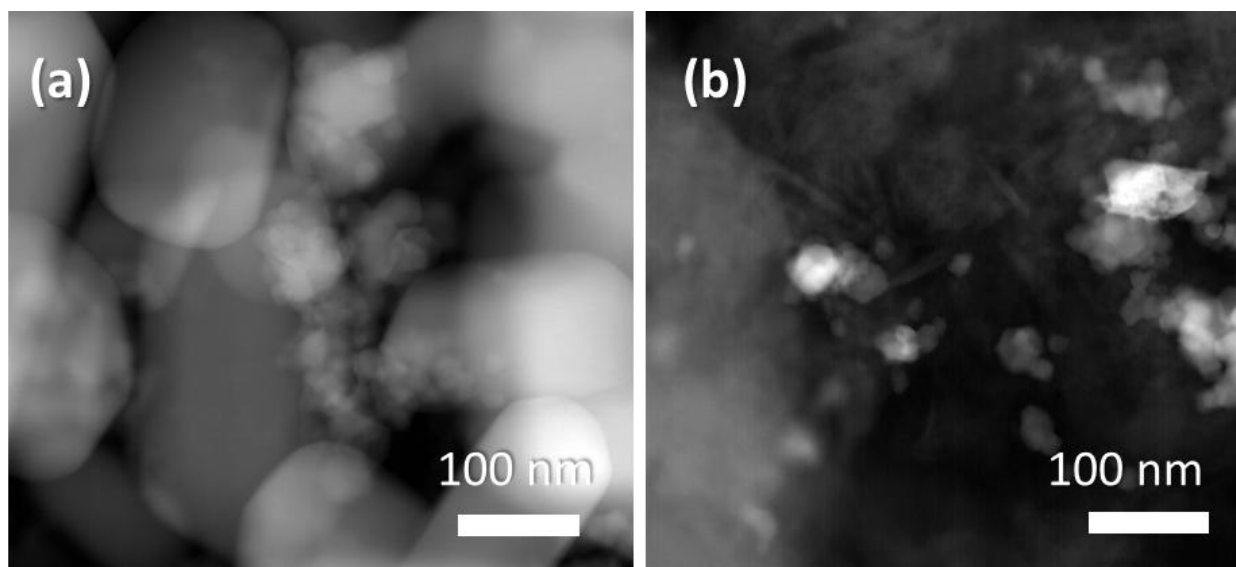


Fig. S1. STEM images of HEO/TiO₂ (a) and HEO/BN (b) materials.

The results of EDX spectroscopy analysis performed for all the samples studied are given in Table S1.

Table S1. Material elemental composition as determined by EDX spectroscopy

Sample	Chemical composition, at.%								
	Cr	Fe	Co	Ni	Mn	O	B	N	Ti
HEO	9.6	9.3	9.4	5.6	8.1	57.9	-	-	-
HEO/BN	1.0	0.9	0.8	0.4	0.7	19.7	45.4	31.1	-
HEO/TiO ₂	0.3	0.3	0.3	0.2	0.2	74.0	-	-	24.7

EDX spectra of HEO (a), HEO/BN (b) and HEO/TiO₂ are collected in Figure S2.

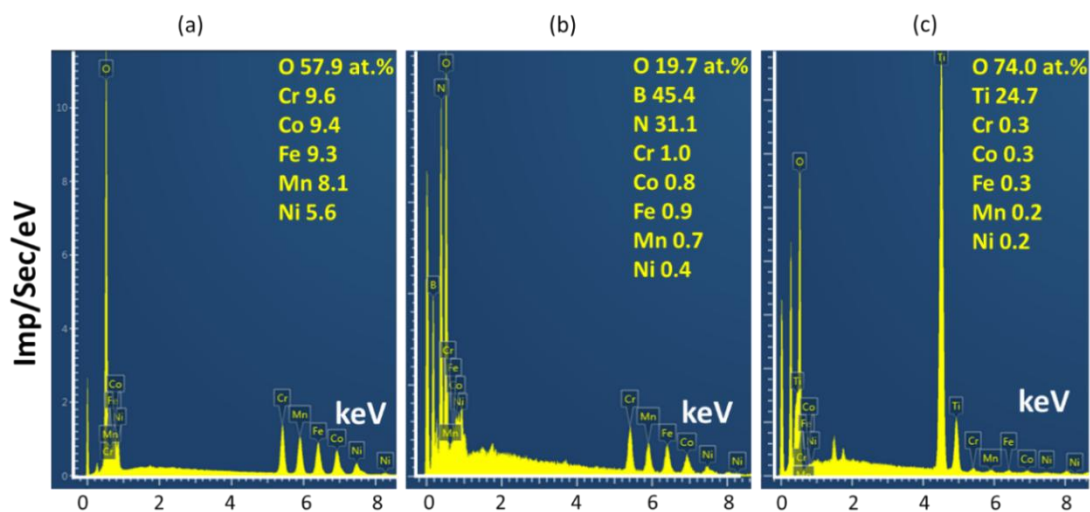


Fig. S2. EDX spectra of HEO (a), HEO/BN (b) and HEO/TiO₂ (c) samples.

DTGA curves allow us to investigate mass loss rate of all samples at higher temperature (Figure S3). The reduction process is most pronounced in the curve of the freestanding HEO NPs. For the curves corresponding to the supported samples, the mass loss is less pronounced but still noticeable compared to the carrier curves. It should be noted that the aliquots of the supported samples contained 4.3 mg of HEO, whereas the bare HEO sample weighed 34.7 mg, eight times more. No mass loss was observed for the pure carrier samples in this temperature range.

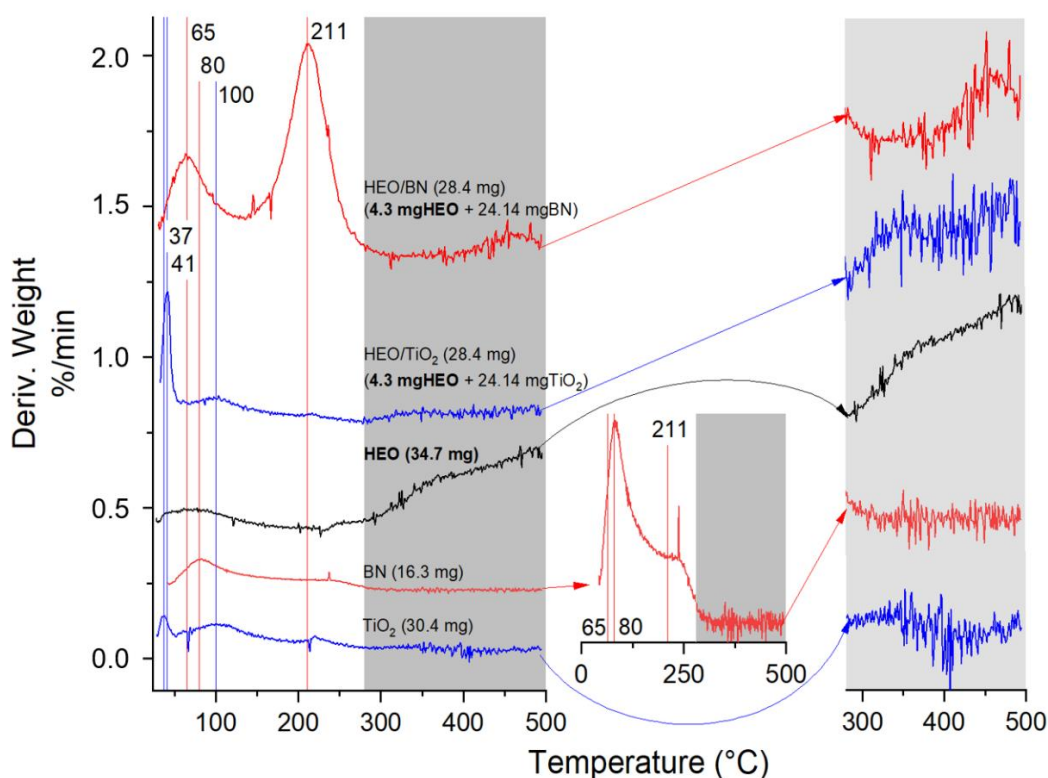


Fig. S3. DTGA curves analysis for all studied materials.

Table S2 compares HEO NPs with other catalytic materials in the CO₂ conversion reaction. However, it is worth noting that the reaction conditions differ both in terms of pressure values, which largely determine the reaction mechanism and material selectivity, and in the H₂/CO₂ ratio, which significantly affects the conversion value. The HEO NPs show an exceptionally high CO₂ conversion at H₂/CO₂ = 2. This is especially noticeable when compared to other multicomponent systems with comparable or lower conversion at higher H₂/CO₂ ratio. The H₂/CO₂ ratio of 2 is not stoichiometric, which leads to hydrogen deficiency. Therefore, the conversion values of 46% is in fact close to the highest possible conversion in the given conditions.

Table S2. Comparison of different catalysts in the CO₂ hydrogenation/methanation reactions

Catalyst	T _{act} , °C	H ₂ /CO ₂	P, MPa	T, °C	X (CO ₂), %	VHSV, L* _{g_{cat}} ⁻¹ *h ⁻¹	Ref
(CrFeCoNiMn) ₃ O ₄	500	2	2	380	46	10.88	This work
Fe _x Ni ₂ CrCoMnO _y	600	4	0.1	400	73	9	[1]
Fe-Co/K-Al ₂ O ₃	400	3	2	320	49	3	[2]
Co ₃ MnNiCuZnO _x	600	3	0.1	500	47	20	[3]
10Ni/Ce-ZrO ₂	450	4	0.1	275	55	20	[4]
10% Ni/TiO ₂	400	4	0.1	350	53.5	60	[5]
0.03% Pt-20% Co-80% Al ₂ O ₃	400	4	0.1	400	70	36	[6]
Ni/CaO-Al ₂ O ₃	450	4	0.1	400	81	48	[7]
CuO-ZnO-TiO ₂ -ZrO ₂	300	3	3	240	43.8	2.4	[8]
Fe-K/γ-Al ₂ O ₃	450	3	1	300	30.9	1.8	[9]
Fe-Cu-K-Al	450	3	0.1	300	39.4	2	[10]
5Mn-Na/Fe	350	3	3	320	38.6	2.04	[11]
La _{0.9} Sr _{0.1} CuO	300	2.85	3	300	9.18	8.6	[12]
10Mn-Fe ₃ O ₄	400	3	2	350	44.7	4	[13]
5% Pt/(NiMgCuZnCo)O	500	3.3	0.1	500	47.8	8.66	[14]

The crystal structure and phase composition of all samples subjected to the activation were studied by XRD. The samples were activated, cooled to room temperature and then transferred to the XRD setup. Note that this leads to an inevitable passivation of the reduced samples. The changes in the phase composition are clearly visible for the freestanding HEO NPs (Figure S4a), less noticeable for the HEO/BN sample (Figure S4b) and indistinguishable for HEO/TiO₂ material (Figure S5) due to the low content of the active phase and the overlap of the HEO/substrate reflections. However, after 400 °C activation, no noticeable changes in the HEO phase composition were observed. The possibility that no reduction occurred at this temperature can be ruled out since the same catalytic efficiency was observed for the HEO samples. It can therefore be assumed that some HEO was reduced to the metallic state and then passivated in air. However, in the case of HEO NPs activated at 500 °C, a set of (111), (200) and (220) peaks of a multicomponent alloy are observed. Apparently, most of the HEO was reduced and air passivation did not lead to complete oxidation. The XRD patterns of HEO NPs activated at 600 °C and 500 °C are similar.

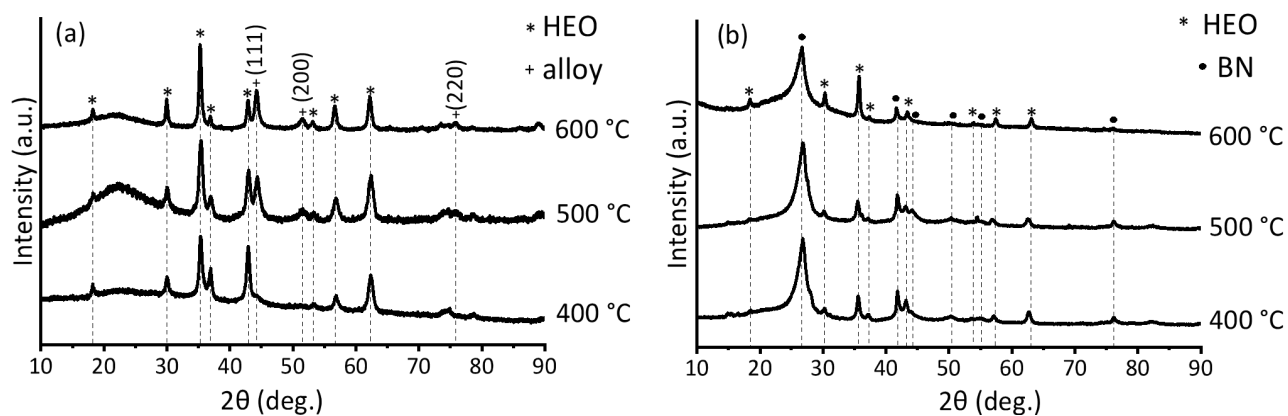


Fig. S4. XRD patterns of HEO (a) and HEO/BN (b) samples after activation at 400°C, 500°C and 600°C.

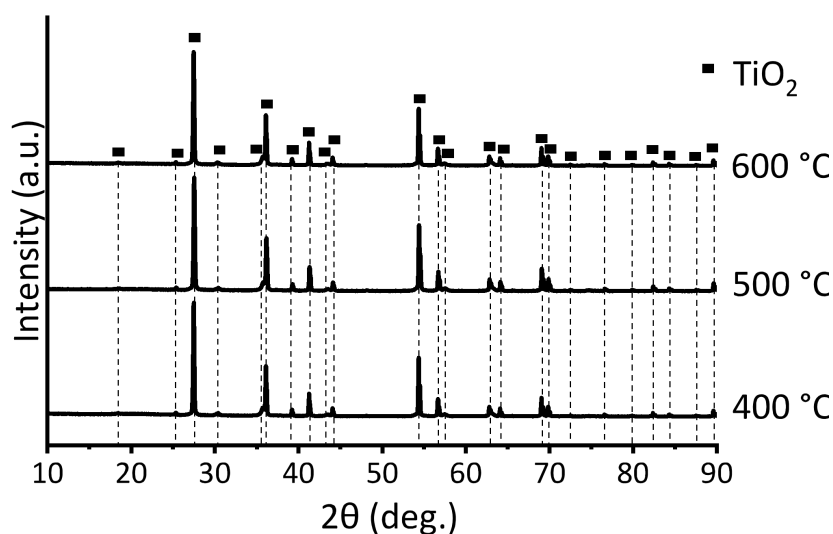


Fig. S5. XRD patterns of HEO/TiO₂ samples after activation at 400°C, 500°C and 600°C.

The X-ray diffraction results for catalysts after conversion tests are shown in Figure S6.

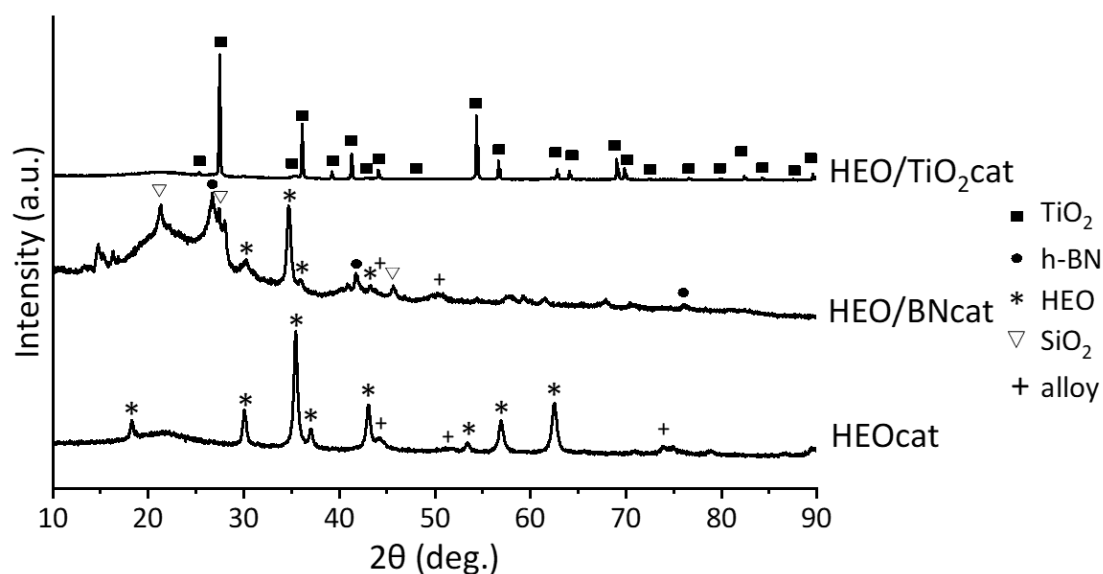


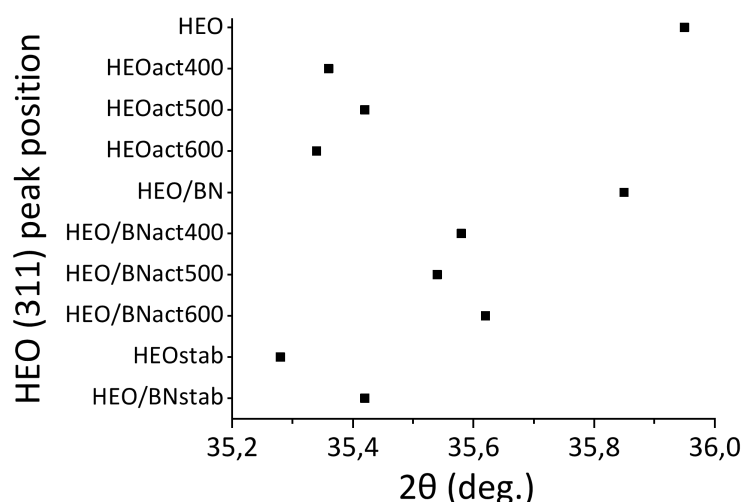
Fig. S6. XRD patterns of HEO, HEO/BN and HEO/TiO₂ samples after catalysis

A comparison of the reaction rate of HEO NPs with some other catalysts is given in Table S3.

Table S3. Reaction rates of various catalysts in CO₂ hydrogenation reaction

Catalyst	T, °C	CO ₂ activity $\mu\text{mol/g}\cdot\text{s}$	Ref
(CrFeCoNiMn) ₃ O ₄	380	6.68	This work
NiO/ZnAl ₂ O ₄	450	10.6	[15]
0.08wt%Na/ZnFe ₂ O ₄	340	1.807	[16]
Co ₃ O ₄	450	3.335	[17]
Cu _x Zn _{1-x} Al ₂ O ₄	250	0.687	[18]
3.5 wt% Rh/TiO ₂	500	13.6	[19]
Co ₃ MnNiCuZnO _x	500	0.003	[3]
5% Pt/(NiMgCuZnCo)O	500	0.012	[14]
ZnFeOx-nNa	320	8.827	[20]

Note the observed systematic shift of the (311) HEO XRD peak towards lower 2θ values, both in the case of activated samples and after their catalytic stability tests (Figure S7), which reaches 0.5–0.7° 2θ for the HEO NPs. In the case of the HEO/BN material, the shift is only 0.3 – 0.4° 2θ . It can be assumed that this shift is associated with the lattice deformation arising from the reduction/oxidation cycles and the corresponding alloy and oxide formation. This shift can be used as a measure of the reduction completeness, which is higher for HEO NPs and lower for HEO/h-BN counterparts. This is another indication that h-BN hinders the HEO reduction.

**Fig. S7.** XRD (311) peak position (2θ) of HEO sample.

The XPS method was used to analyze the surface chemical states of the studied materials. Symmetric peak shapes, typically Voigt profiles, are employed for insulating compounds like most metal oxides. This is because the absence of accessible electronic states at the Fermi level prevents low-energy electron-hole pair excitations, resulting in a symmetrical line shape. Conversely, asymmetric line shapes are necessary for metallic species and certain conducting oxides, where the high density of states at the Fermi level enables screening of the core hole via these excitations, producing a characteristic tail on the high-binding-energy side of the peak [21].

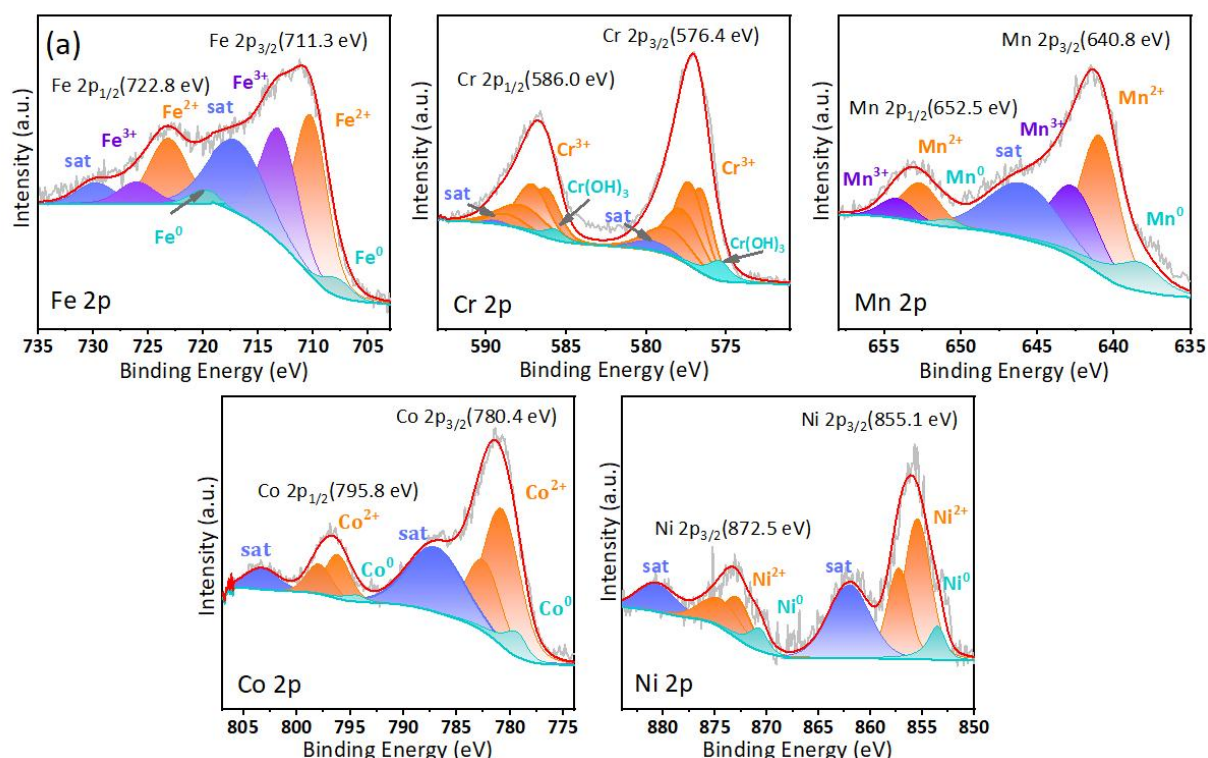
Based on the well-pronounced satellite structure, it is deduced that Co is present as Co²⁺ [22,23]. Please note that despite being represented by a single oxidation state, the main Co 2p peak is approximated by two components. This is due to the final state relaxation effect, which results in two distinct final states upon photoemission. After conversion testing (Figure S8) a peak at 778.4 eV appears, corresponding to metallic Co [24]. Similarly, Ni is also represented as Ni²⁺ ions [22,23,25]. Cr is in the Cr³⁺ oxidation state and is approximated by a multiplet, as

suggested elsewhere [23]. Based on these findings, the following conclusions can be drawn. According to crystal field theory, Cr^{3+} ions are expected to preferentially occupy octahedral sites [26] due to their high crystal field stabilization energy (CFSE). In the HEO sample after conversion testing, a noticeable shift towards lower BE values is observed (Figure S8). The $\text{Cr}2p_{3/2}$ and $\text{Cr}2p_{1/2}$ maxima are located at 576.37 eV and 586.0 eV, respectively. The $\text{Cr}2p_{3/2}$ maximum reveals a new contribution at 575.7 eV, which can be assigned to Cr^{3+} in $\text{Cr}(\text{OH})_3$ [27].

Similarly, both Ni^{2+} and Co^{2+} also exhibit high CFSE, suggesting their preference for octahedral coordination, analogous to the cation distribution in an inverse spinel. Therefore, the obtained spinel is neither purely normal nor purely inverse but possesses a complex cation distribution [28].

In the as-synthesized HEO sample, the XPS $\text{Mn}2p$ spectrum shows a characteristic doublet at positions 641.6 ($\text{Mn}2p_{3/2}$) and 653.2 eV ($\text{Mn}2p_{1/2}$) [29], which were deconvoluted using three components. The peak at 640.65 eV and a distinct satellite at 645.4 eV indicate the presence of Mn^{2+} species [30], and the maximum at 642.4 eV can be attributed to Mn^{3+} [31]. After conversion testing (Figure S8) an additional component appears at 639.0 eV, which corresponds well to the Mn^0 state [32].

The XPS $\text{Fe}2p$ spectrum of the HEO sample was fitted with two components, revealing the presence of Fe^{2+} (710.2 eV) and Fe^{3+} (713.1), which is typical for spinel oxide [33]. In the sample after conversion, the positions of the doublet peaks remain almost unchanged. An additional small maximum at 707.9 eV, observed after spectrum deconvolution, is characteristic of the metallic state of iron [34].



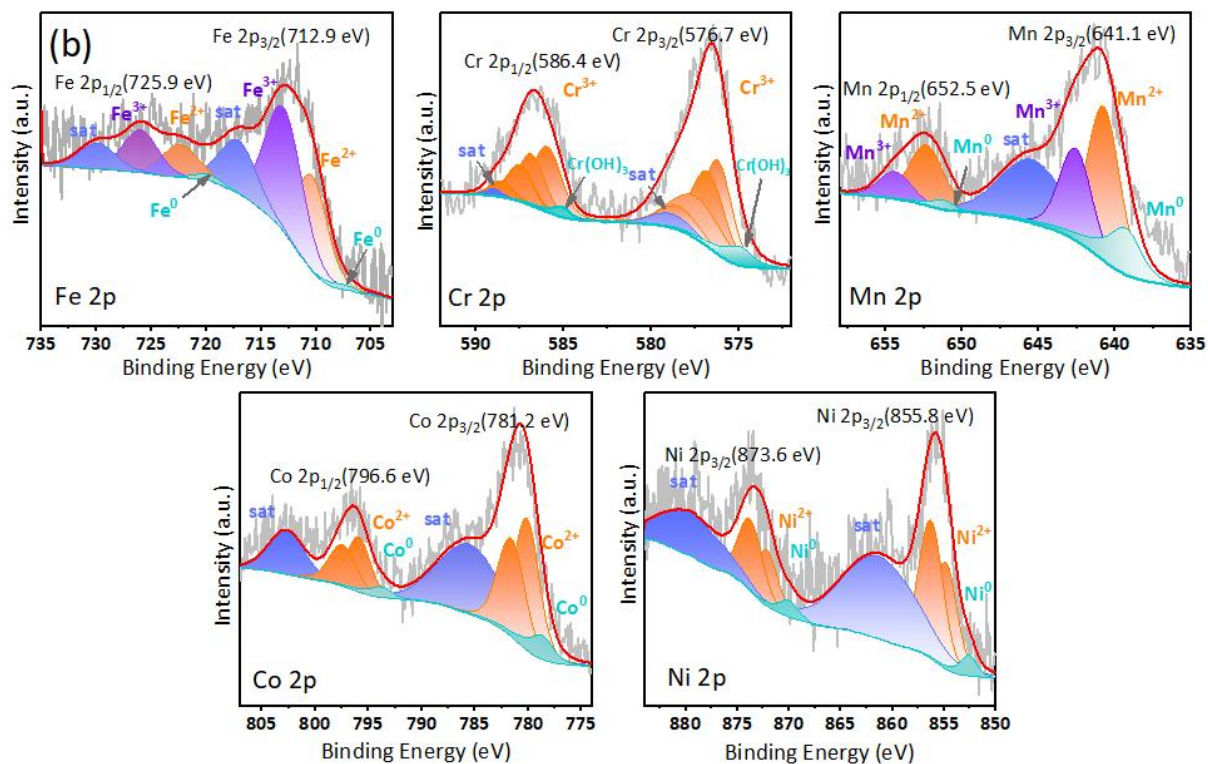


Fig. S8. XPS spectra of HEO materials after conversion (a) and stability (b) tests.

The XPS spectra for supported HEO/TiO₂ and HEO/h-BN samples after stability testing are shown in Figures 9 and 10.

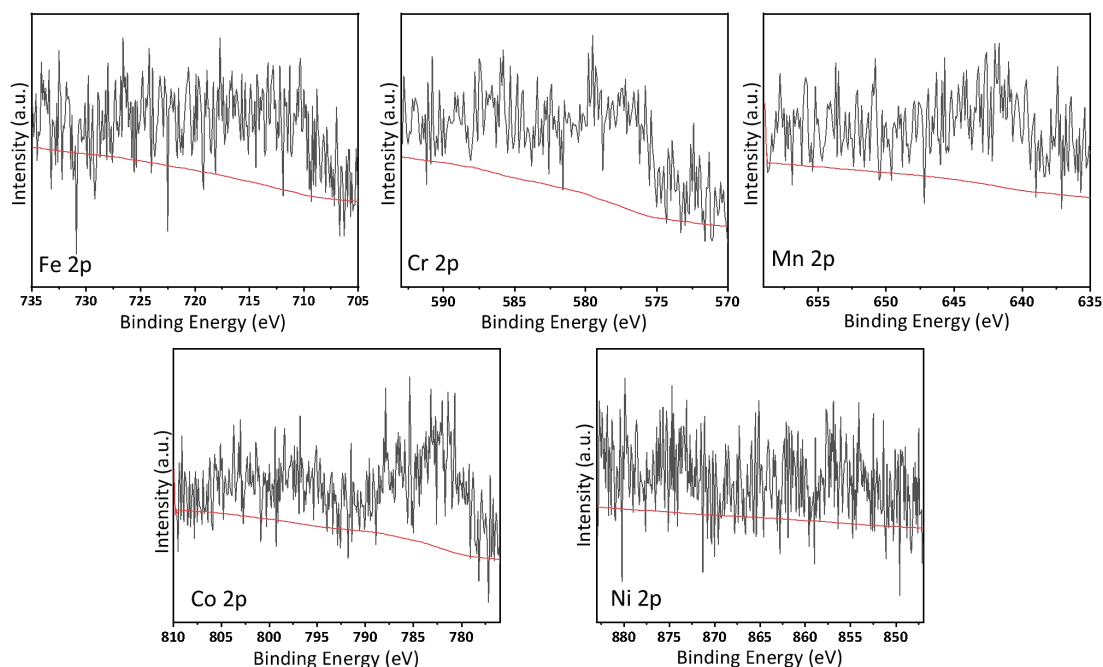


Fig. S9. XPS spectra of HEO/TiO₂ material after stability test.

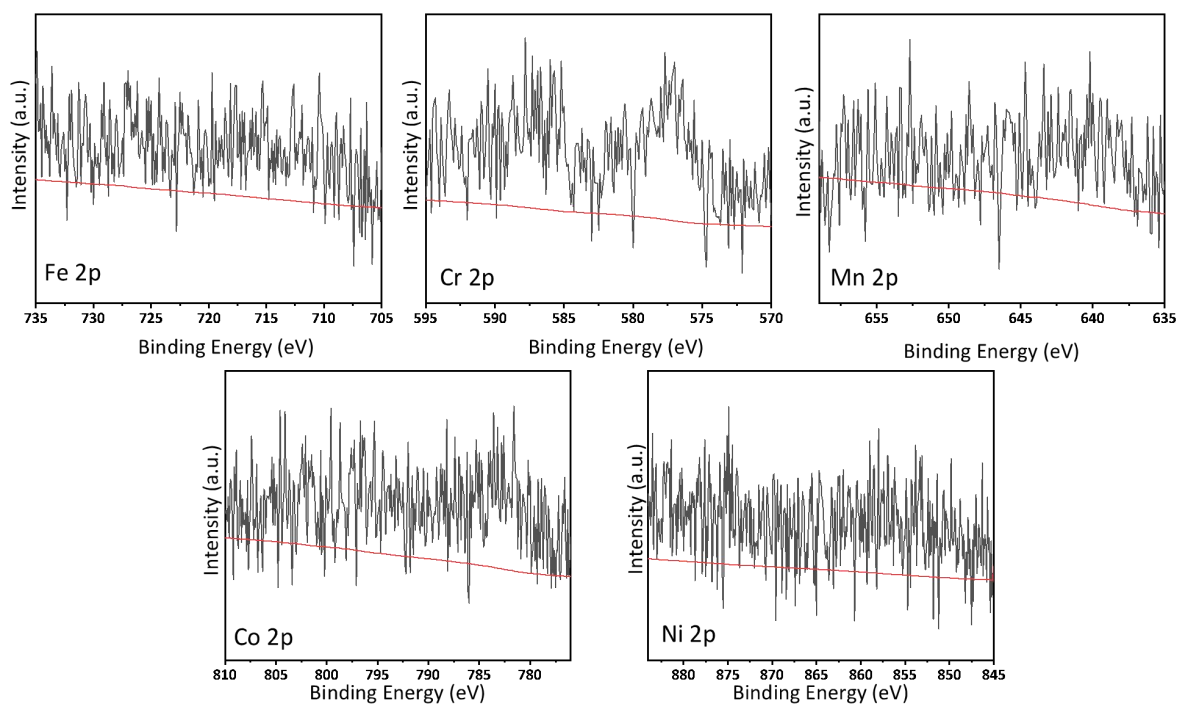


Fig. S10. XPS spectra of HEO/BN material after stability test.

Table S4 presents HEO NP surface composition in as-synthesized state and after catalytic conversion and stability tests.

Table S4. HEO NP surface composition in as-synthesized state and after catalytic conversion and stability tests.

Sample	Concentration, at.%				
	Cr	Co	Fe	Mn	Ni
HEO NPs as-synthesized	24.4	26.6	24.8	22	2.2
HEO NPs after conversion testing	26.7	30	28.3	12	3
HEO NPs after stability testing	33.8	23.7	12	17.5	13

The X-ray diffraction results for catalysts after stability tests are shown in Figure S11.

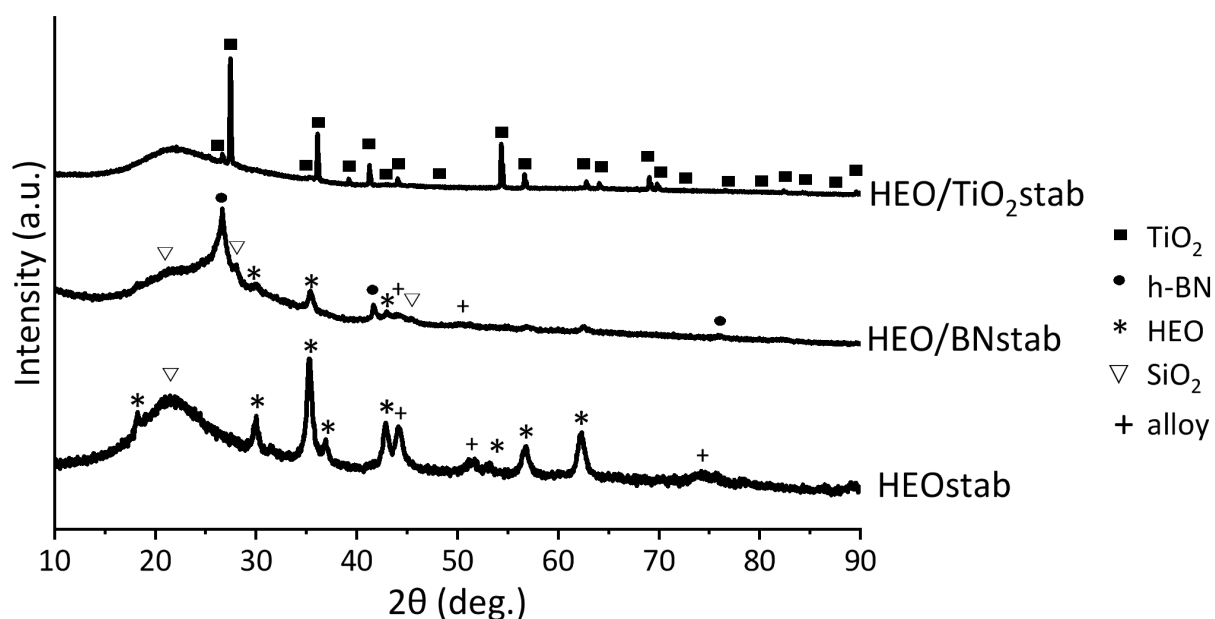


Fig. S11. XRD patterns of HEO, HEO/BN and HEO/TiO₂ samples after stability tests.

STEM images and EDX spectroscopy elemental maps of HEO/BN and HEO/TiO₂ materials after catalytic stability tests are shown in Figures S12 – 14.

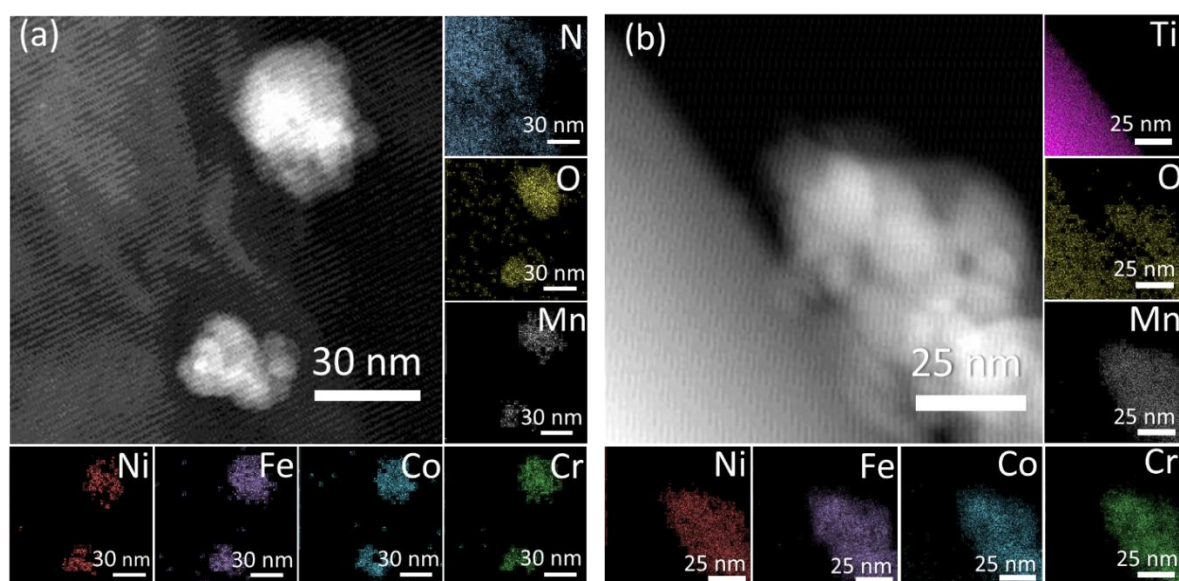


Fig. S12. STEM image and corresponding EDX spectroscopy elemental maps of HEO/BN (a) and HEO/TiO₂ (b) materials after catalytic stability tests. The distribution of boron (B) in the HEO/BN sample is the same as the distribution of nitrogen (N) and is not shown here.

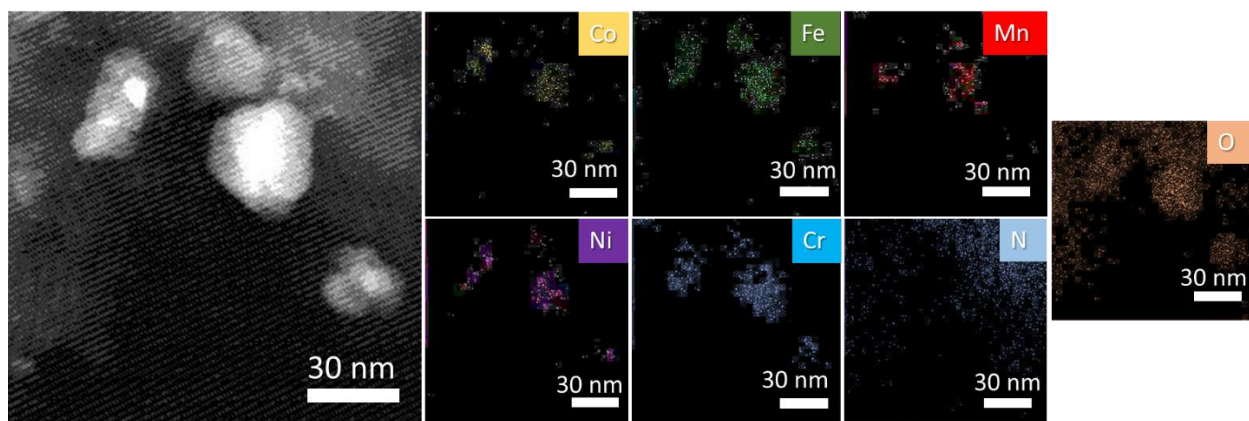


Fig. S13. STEM image and corresponding EDX spectroscopy elemental maps of HEO/BN after catalytic stability tests. The distribution of boron (B) in the HEO/BN sample is similar to that of nitrogen (N) and is not shown here.

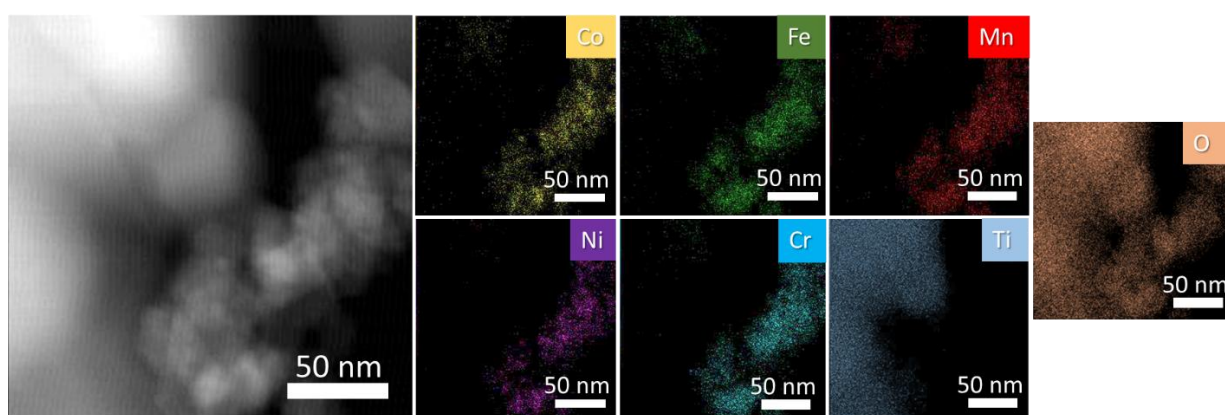


Fig. S14. STEM image and corresponding EDX spectroscopy elemental maps of HEO/TiO₂ materials after catalytic stability tests.

Consequent TPR/TPO profiles is shown in Figure S15.

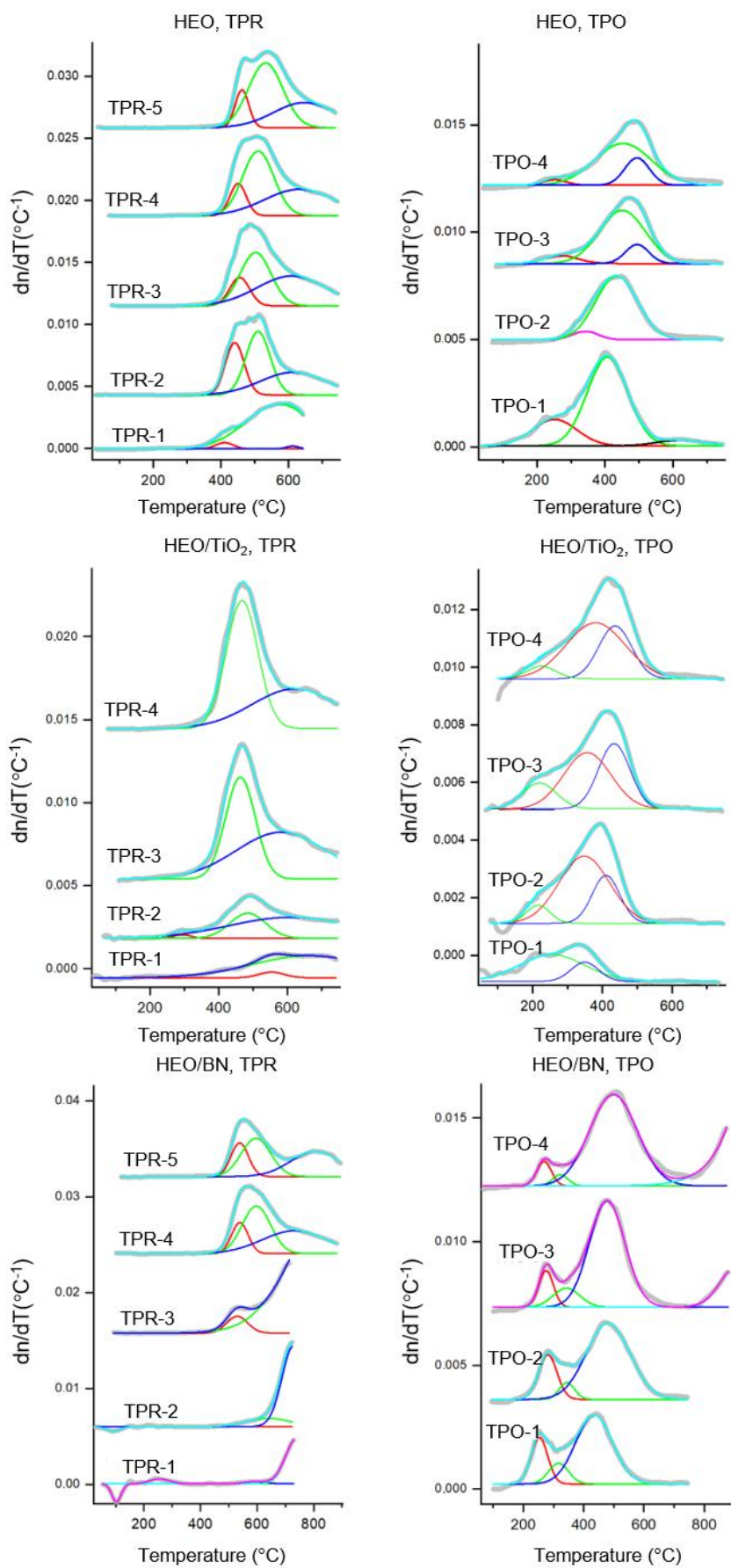


Fig. S15. Consequent TPR/TPO runs.

Comparison of initial TPR curves obtained during first runs are given in Figure S16. For all samples, negative TPR and TGA peaks are observed in the low-temperature region, supporting the hypothesis of water release. TPR and TGA peaks in the 200–300 °C range indicate hydrogen uptake during the reduction of BNO bonds. In TGA profiles, the reduction of HEO is weakly pronounced, whereas in TPR profiles, this process remains clearly detectable up to 650–750 °C. The observed shift of TPR peaks to higher temperatures (by approximately 30 °C) compared to corresponding TGA peaks is attributed to differences in experimental conditions.

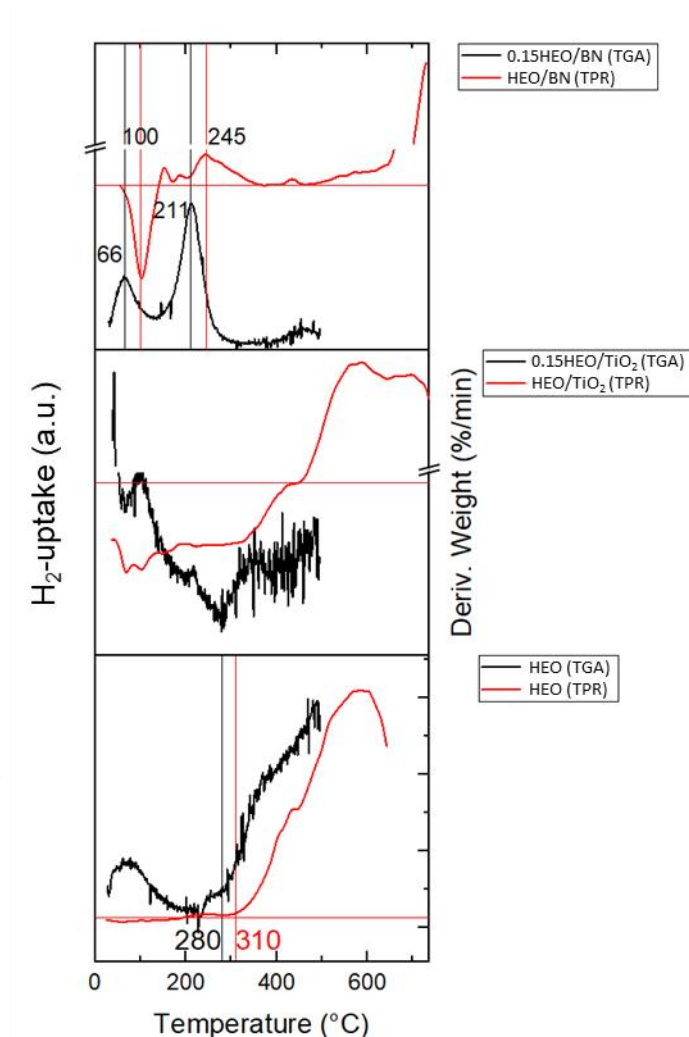


Fig. S16. Comparison of TRP and TGA curves.

Figure S17 collects different H_2/O_2 ratios for HEO NPs, HEO/TiO₂ and HEO/BN samples after consequent TPR/TPO runs. For the HEO and HEO/TiO₂ samples, the H_2/O_2 ratio approaches 2 (corresponding to water formation stoichiometry) starting from the 2nd or 3rd cycle. This indicates the reversibility of the redox processes, which we interpret as reversible transitions between HEO and HEA. In the case of the HEO/BN sample, the H_2/O_2 ratio increases with each subsequent cycle but does not reach 2 within 4 cycles. This suggests an irreversible oxidation process, that could be associated with a constant hydrogen depletion due to adsorption by h-BN. An increase in the maximum temperature, starting from the third oxidation cycle, leads to the following effects: (i) the emergence of an oxygen uptake region above 750 °C; (ii) a decrease in the onset temperature of hydrogen absorption to approximately 450 °C, while the profile shape approaches that characteristic of the HEO and HEO/TiO₂ samples; (iii) a more

rapid increase in the H_2/O_2 ratio. It can be assumed that at a higher temperature and a greater number of TPR/TPO runs, h-BN is saturated with hydrogen and the reduction of HEO NPs becomes more pronounced.

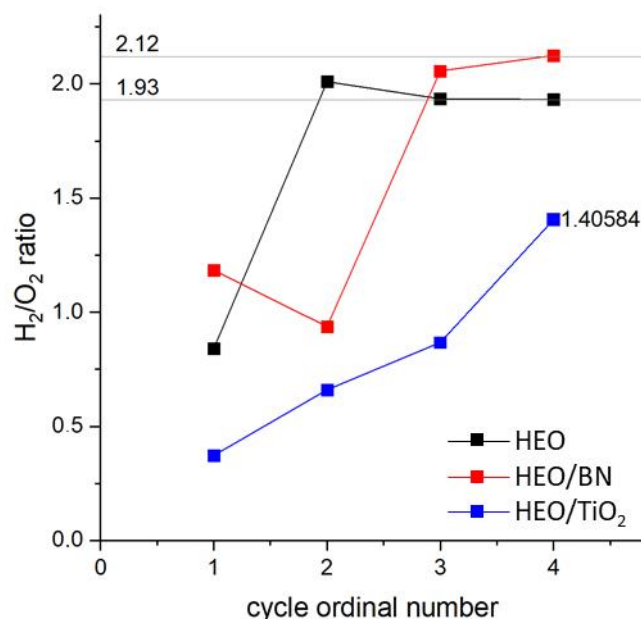


Fig. S17. H_2/O_2 ratios for HEO NPs, HEO/TiO₂ and HEO/BN samples during consequent TPR/TPO runs.

A direct comparison of the TPR profiles of the HEO sample with literature data was performed (Figure S18). The HEO sample was selected for this comparison to eliminate possible influence of HEO-support interactions on the profile shape. When selecting literature data, special attention was paid to studies that present TPR spectra for unsupported oxides. The list of references is presented in Table S5.

The vertical lines in Figure S18 indicate the peak maximum positions in the TPR profiles. The color of each vertical line corresponds to the color of the reference number. Before analyzing the data, it should be noted that oxide reduction typically occurs in several steps ($Me^{n+} \rightarrow Me^{m+} \rightarrow Me^{k+} \rightarrow \dots$, where $n > m > k$, $k \geq 0$), with each reduction step producing a characteristic peak in the TPR profile, although these peaks are not necessarily fully resolved.

Comparison of the experimental data with literature sources does not reveal absorption in the regions characteristic of chromium[35,36] and nickel[37,38] oxides, as well as in the regions corresponding to higher oxidation states of cobalt, iron, and manganese [39–52]. Thus, a complete absence of absorption is observed in the regions typical for two of the five HEO metals, as well as in the regions characteristic of high oxidation states of the remaining three metals. This additionally confirms both the absence of separate individual oxide phases and the formation of HEO with properties that differ from the sum of the properties of its constituent oxides.

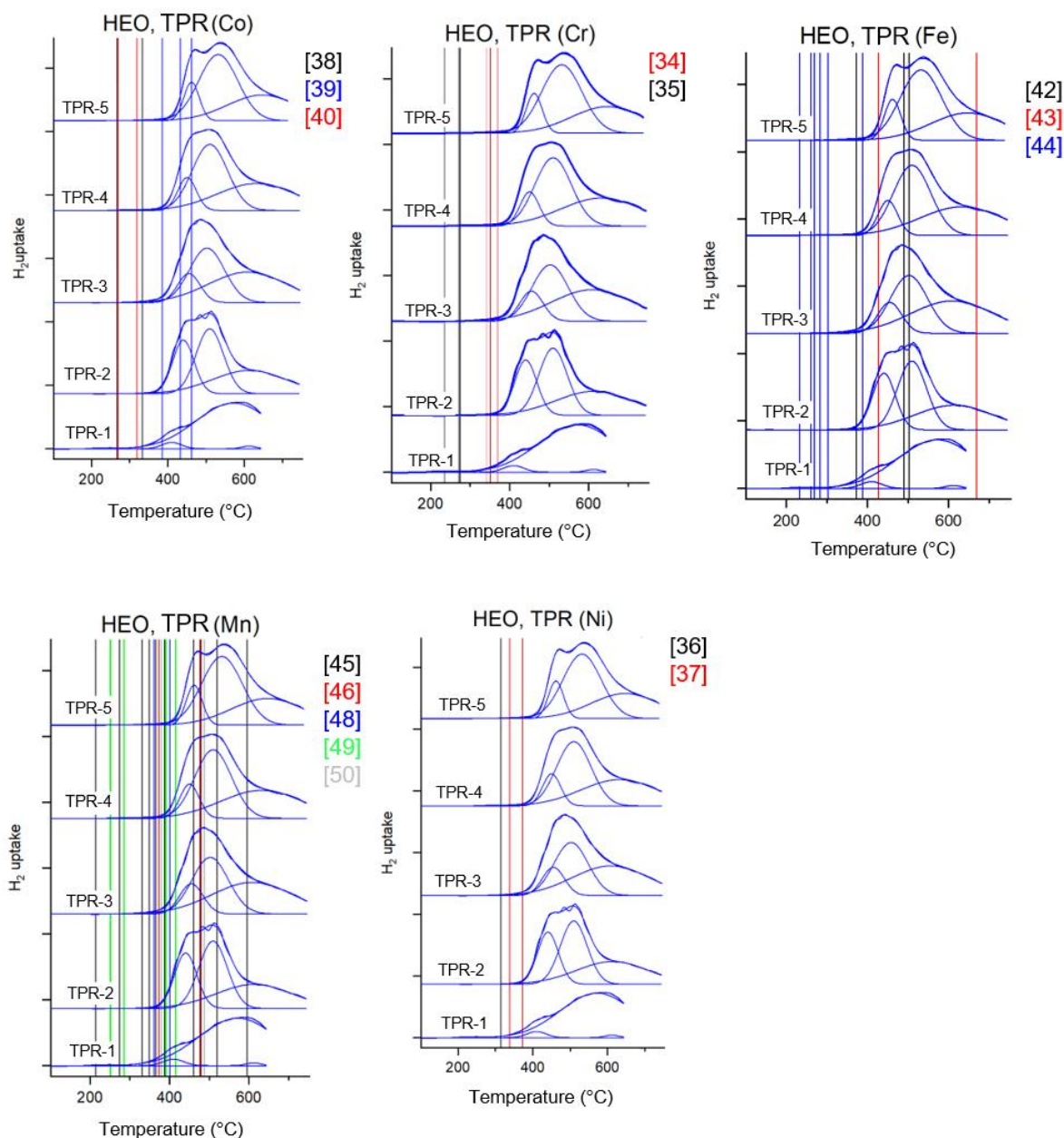


Fig. S18. Comparison of the obtained TPR profiles of HEO NPs with typical TPR peak positions of Co, Cr, Fe, Mn and Ni oxides according to literature data.

Typical TPR peak positions of Co, Cr, Fe, Mn and Ni oxides according to literature data are given in Table S5.

Table S5. Typical TPR peak positions of Co, Cr, Fe, Mn and Ni oxides according to literature data.

No	metal	oxide	support	T _m , °C	comments	Ref.
1.	Co	Co ₃ O ₄	-	268 334		[39]
2.	Co	Co ₂ O ₃	-	386 433 462		[40]
3.	Co	Co ₃ O ₄	-	270 320		[41]
4.	Cr	no data	ZrO ₂	350 342 369 350	(%Cr = 0,5) (%Cr = 1,0) (%Cr = 4,0) (%Cr = 8,0)	[36]
5.	Cr	no data	-	234 273 274 234 271	(Cr ⁶⁺ →....) (Cr ⁿ⁺ →..., n<6)	[35]
6.	Cr	no data	-	352 268 348	(Cr ⁶⁺ →....) (Cr ³⁺ →....)	[42]
7.	Fe	Fe ₂ O ₃	-	373 491		[43]
8.	Fe	Fe ₃ O ₄ no data	-	503 428 669		[44]
9.	Fe	no data	-	234 270 262 284 389		[45]
10.	Mn	MnO ₂	-	274 331 214 349 461 596 388 366 479 521	for various allotropic modifications of MnO ₂	[46]
11.	Mn	Mn ₂ O ₃	-	374 477		[47]
12.	Mn	Mn ₂ O ₃	-	323 417		[48]
13.	Mn		-	550		[41]
14.	Mn	MnO ₂	-	361		[49]

15.	Mn	MnO ₂	-	401 251 283 285 390 291 303 415 317 337	for various allotropic modifications of MnO ₂	[50]
16.	Mn	Mn ₂ O ₃	-	379 a 434 a 420 b 487 b 408 c 474 c	letters refer to different possible morphologies	[51]
17.	Mn	MnO _x	-	190 °C 328 °C 424 °C 389 °C 451 °C	MnO ₂ → Mn ₂ O ₃ → Mn ₃ O ₄ → MnO	[52]
18.	Ni	NiO	-	315		[38]
19.	Ni	NiO	-	339 373		[37]

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